

INTERFACE STRUCTURE IN TiN/SiC NANOCOMPOSITES

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Extended experience in the design of super-hard nanocomposites and understanding of their properties has been achieved during the last decade. TiN/SiN_x nano-composites are the most studied systems [1]. The strong increase in hardness as compared with pure TiN has been attributed to the nanometer scale randomly oriented TiN grains and one monolayer thick (1 ML) SiN_x interfacial layer [1]. The TiN/SiC hetero-structures have been studied to a lesser extent. Kong, Dai, Lao *et al.* have deposited nanolayered TiN/SiC coatings at room temperature [2]. They showed that the 3C-SiC layers have crystallized and grown with TiN layers in the (111) growth orientation.

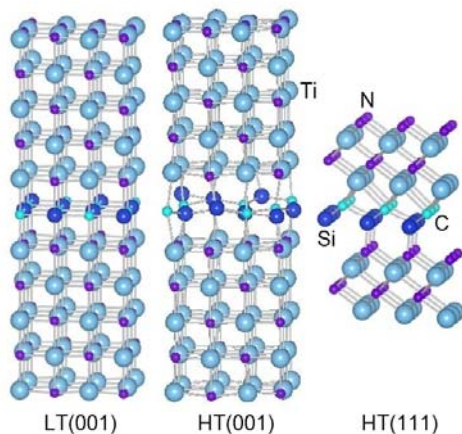


Fig.1. Atomic configuration of the low- (LT) and high-temperature (HT) TiN/1 ML SiC/TiN hetero-structures

In this work, we investigate the hetero-structures with one monolayer of interfacial SiC inserted between several B1(NaCl)-TiN (001) and (111) monolayers in the temperature range of 0-1400 K using first-principles quantum molecular dynamics (QMD) simulations. The temperature-dependent QMD calculations in combination with subsequent variable-cell structural relaxation reveal that the TiN(001)/B1-SiC/TiN(001) interface exists as pseudo-morphic B1-SiC layer at 0 K and 600 K (LT(001), cf. Fig. 1). After heating to 900-1400 K and subsequent static relaxation (HT(001)), the interfacial layer corresponds to a strongly distorted 3C-SiC-like structure oriented in

the (111) direction in which the Si and C atoms are located in the same interfacial plane (cf. Fig. 1). The Si atoms form the four-fold Si-C₃N₁ coordinated configurations, whereas the C atoms are located in the C-Si₃Ti₂ surrounding.

All the (111) interfaces simulated at 0, 300 (LT(111)) and 1400 K (HT(111)) have the same atomic configurations. For these interfaces, the Si and C layers are similar to those that are aligned perpendicularly to the (111) direction in 3C-SiC. The Si and C atoms are located in Si-C₃N₁ and C-Si₃Ti₆ configurations, respectively.

The B1-SiC → 3C-SiC transformation at the interface of TiN/SiC nanocomposites above 600 K is confirmed with the XRD spectra of the nanolayered TiN/SiC coatings, presented in Fig. 2. We have deposited these coatings by dual magnetron sputtering the TiN and SiC target at different temperatures (T_S). It is clearly seen that the intensity of the SiC reflex at 2θ~35.6° jumps for the coating deposited at T_S above 600 K, in agreement with the results of our calculations.

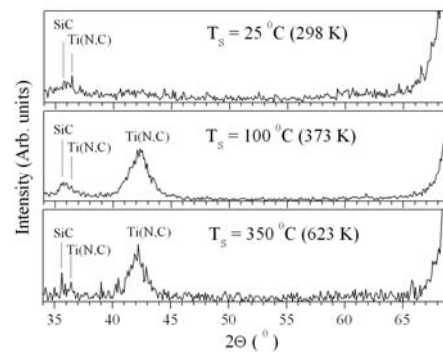


Fig. 2. XRD spectra of TiN/SiC nanolayered coatings. T_S is substrate temperature

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